

# Electrical and Compositional Properties of TaSi<sub>2</sub> Films

N.M. RAVINDRA,<sup>1,4</sup> LEI JIN,<sup>1</sup> DENTCHO IVANOV,<sup>1</sup> VISHAL R. MEHTA,<sup>1</sup>  
LAMINE M. DIENG,<sup>2</sup> GUERMAN POPOV,<sup>3</sup> OKTAY H. GOKCE,<sup>1</sup>  
JAMES GROW,<sup>1</sup> and ANTHONY T. FIORY<sup>1</sup>

1.—New Jersey Institute of Technology, University Heights, Newark, NJ 07102. 2.—Department of Physics and Astronomy, Rutgers University, Piscataway, NJ 08854. 3.—Department of Chemistry, Rutgers University, Piscataway, NJ 08854. 4.—E-mail: nmravindra@comcast.net

Tantalum silicide (TaSi<sub>2</sub>) thin films were sputter deposited on p- and n-type silicon substrates using ultrapure TaSi<sub>2</sub> targets. The TaSi<sub>2</sub>/Si samples were annealed in nitrogen or forming gas or oxygen containing steam at temperatures in the range of 400–900°C. The sheet resistances of TaSi<sub>2</sub>/Si were measured by four-point probe before and after anneal. The structure of these films was investigated using x-ray diffraction (XRD) methods. It has been found that the sheet resistance decreases with the increase in annealing temperature and also with the increase in film thickness. X-ray diffraction patterns show changes in the morphological structure of the films. Oxidation characteristics of the films have been investigated in the temperature range of 400–900°C in oxygen containing steam ambient. The oxidation time ranged from 0.5 to 1.5 h. No oxide formation of the tantalum silicide films was observed in this investigation. This has been attributed to the high purity of TaSi<sub>2</sub> sputter targets used in the preparation of the films.

**Key words:** Tantalum silicide, annealing, sputtering, sheet resistance, oxidation

## INTRODUCTION

With continuously decreasing device dimensions in silicon-integrated circuit technology, the relatively high sheet resistance of doped polycrystalline silicon as the gate metal has become a limiting factor for device performance. This has led to increased interest in incorporating refractory metal silicides into the device fabrication process. The potential benefits of smaller integrated-circuit dimensions can be realized only if measures are taken to prevent parasitic effects from dominating the electrical characteristics. One of the most urgent needs is to find a low-resistivity substitute for heavily doped polycrystalline silicon (poly-Si) at the gate level. Due to the advantages of ease of formation, low resistivity, and thermal stability, TiSi<sub>2</sub> and TaSi<sub>2</sub> are particularly promising to silicon device technology.

Silicides are most commonly used in dynamic ran-

dom access memory (DRAM) and complementary metal oxide semiconductor (CMOS) devices as conducting layers that can withstand high temperature and that have a significantly low sheet resistivity. Besides their role as a stable conductor, silicides have other interesting applications and properties. These include their use as a diffusion source for impurities to generate shallow junctions, or as an alternative to self-aligned silicides. A selective chemical vapor deposition (CVD) process is interesting because it uses no silicon consumption. Implantation into silicide or even an in-situ doped silicide deposition can eliminate implantation damage in silicon and reduce defects.

Among the metal silicides, TaSi<sub>2</sub> has high-temperature stability and low contact resistance. The most interesting aspect of TaSi<sub>2</sub> is that it can be directly used as a gate material without the need for poly-Si.<sup>1</sup> A large number of studies on TaSi<sub>2</sub> have been published in the literature. In this work, direct current (DC)-sputtering methods have been used to

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sputter deposit TaSi<sub>2</sub> thin films on silicon substrates. The effects of annealing of the films in inert and steam ambient have been investigated. Sheet resistance and x-ray diffraction (XRD) methods have been used to characterize the films.

### EXPERIMENTAL DETAILS

Silicon wafers of orientation (100) type p and n, resistivity 5–10 Ω·cm, 4-in. diameter, and 600-μm thickness have been used as substrate materials. These wafers were cleaned using the standard RCA clean method. A Varian (Peabody, MA) 3125 magnetron DC sputtering system with ultrahigh purity TaSi<sub>2</sub> target was used to deposit tantalum silicide thin films. The sputter deposition chamber was pumped to a pressure of better than  $8 \times 10^{-7}$  Torr. Sputtering was performed at an argon pressure of 5–7 mTorr. Tantalum silicide thin films deposited on silicon wafer had thickness values in the range of 100–1000 Å. The films were annealed in N<sub>2</sub> or forming gas or oxygen containing steam at temperatures of 400–900°C for a duration of 0.5–1.5 h.

### RESULTS AND DISCUSSION

#### Four-Point Probe

The sheet resistance of sputter-deposited films before annealing is listed in Table I. As is shown in the table, the sheet resistance decreases with the increase in thickness of TaSi<sub>2</sub>. The sheet resistance ( $R_s$ ) of film deposited on p-type wafer is higher than that deposited on n-type wafer for the same thickness. The 1000-Å TaSi<sub>2</sub>/Si samples were annealed from 400 °C to 900°C for 30 min in N<sub>2</sub> and the sheet resistance of the films was measured for each annealing temperature. These results are presented in Fig.1. From this figure, it may be noted that the sheet resistance of the TaSi<sub>2</sub> thin film decreases with increasing annealing temperature. A comparison with the literature<sup>2–5</sup> shows that our results are in agreement with the published data.<sup>2</sup> Figure 2 shows the experimental data of the sheet resistance of TaSi<sub>2</sub> films from the literature.<sup>2</sup> TaSi<sub>2</sub> is known to have a barrier height of 0.58 eV on n-Si.<sup>1</sup>

#### X-ray Diffraction

TaSi<sub>2</sub> thin films annealed in forming gas N<sub>2</sub> or oxygen containing steam ambient at various temperatures were analyzed by XRD. The XRD spectra

**Table I. Sheet Resistance of Sputter-Deposited TaSi<sub>2</sub>/Si**

Wafer Type	Film Thickness (Å)	$R_s$ (Ω/Sq)
n	200	66.0
n	600	36.7
n	1000	22.5
p	200	130.1
p	600	46.9
p	1000	27.5

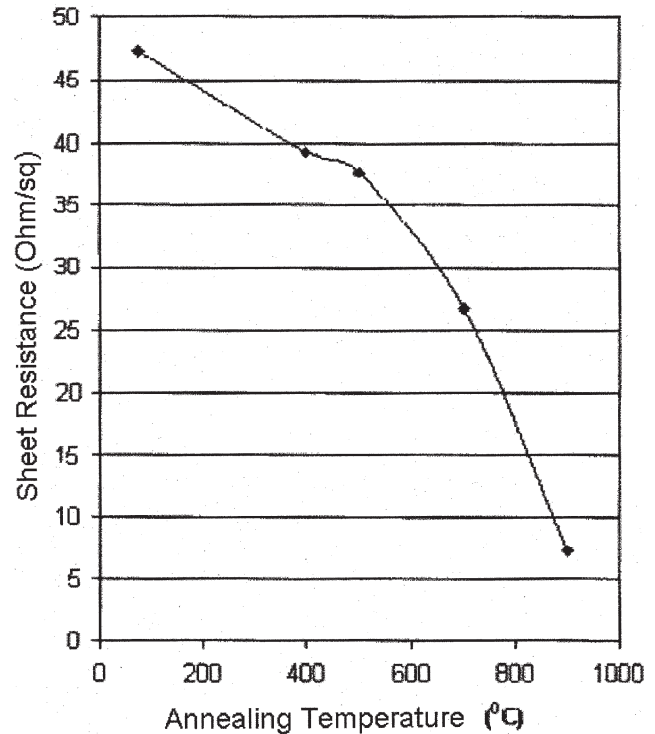


Fig. 1. Sheet resistance of TaSi<sub>2</sub> thin films versus annealing temperature. Film thickness = 1000 Å.

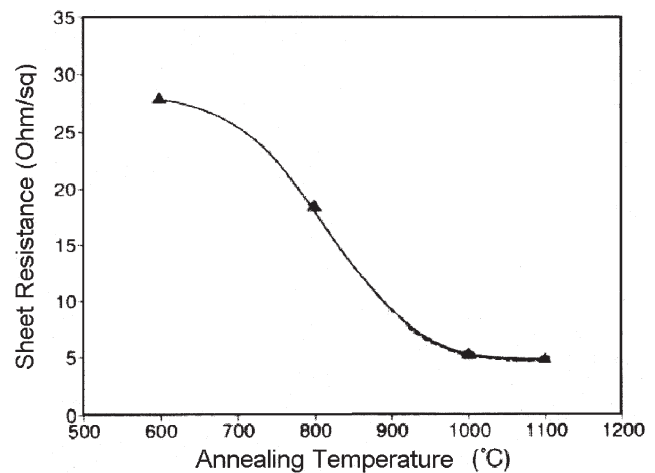


Fig. 2. Cho's experimental data of sheet resistance of TaSi<sub>2</sub> thin films.<sup>2</sup>

of as-deposited TaSi<sub>2</sub> and films annealed at temperatures in the range of 600°C to 900°C are shown in Figs. 3–5. Four sharp features may be observed in the XRD of as-deposited films in Fig. 3. From the XRD results shown in Fig. 4a and b, annealing in the various ambients did not produce significant differences in their influence on the composition of TaSi<sub>2</sub>. From these spectra, it is clear that peaks in addition to those of Si are observed. A comparison with the standard XRD data of TaSi<sub>2</sub><sup>6</sup> confirms that the peaks obtained are due to TaSi<sub>2</sub> (100), (101), (102), (110), (111), (003), (200), and (112) orientations. Other phases such as Ta<sub>2</sub>Si and Ta<sub>5</sub>Si<sub>3</sub> were absent within the detection limit of the experiment.

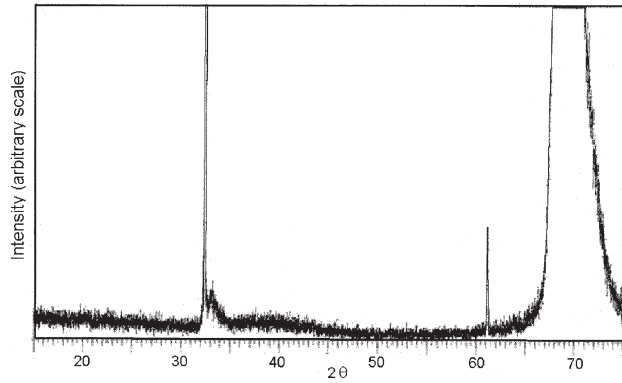
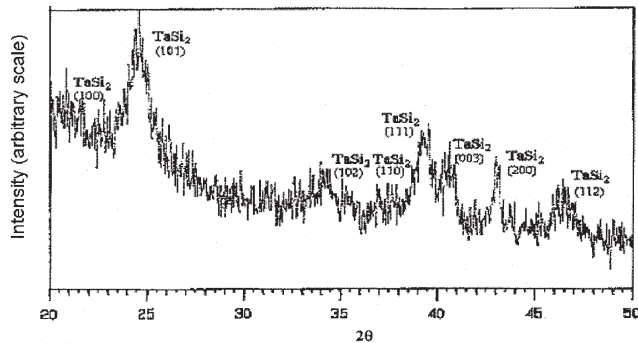
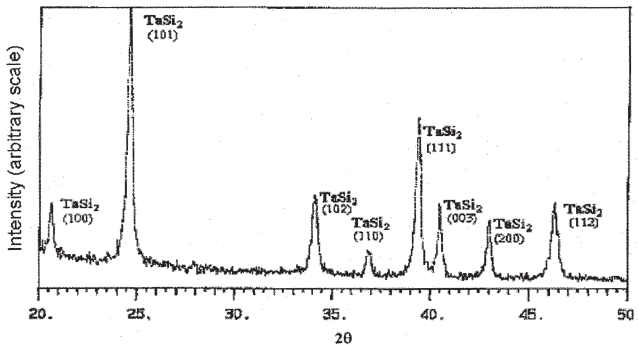


Fig. 3. XRD spectra of as-deposited TaSi<sub>2</sub> thin film (1000 Å).



a



b

Fig. 4. XRD spectra after oxidation of 1000 TaSi<sub>2</sub> thin film on n-Si after annealing: (a) 600°C annealing in N<sub>2</sub> and (b) 900°C annealing in forming gas.

This result agrees well with the XRD data published<sup>3,7</sup> in the literature. Comparing the two XRD data, the peaks in Fig. 4a are broader than those in Fig. 4b. This implies that the microstructures of TaSi<sub>2</sub> thin film become crystallized after annealing and become more ordered with an increase in annealing temperature. From Fig. 5, it can be seen that as annealing temperature increases, the peaks of TaSi<sub>2</sub> thin film become narrower and show an increase in intensity. The data of intensity and full-width half-maximum (FWHM) of the peak (101) are listed in Table II for a quantitative comparison. From this table, it can be seen that between 800°C and 900°C annealing temperatures, the diffraction peak changes abruptly. It can be concluded from this result that the crystallization of TaSi<sub>2</sub> occurs mainly

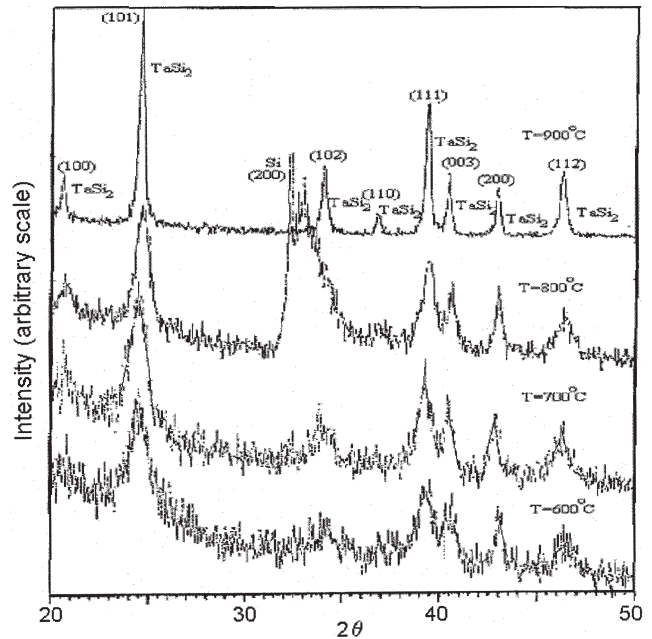


Fig. 5. Comparison of XRD spectra of 1000 Å TaSi<sub>2</sub> thin films on n-Si after annealing (from 600°C to 900°C in steam).

between 800°C and 900°C. Results of Cho et al.<sup>3</sup> in Fig. 6 show the occurrence of a continuous phase transformation above 800°C. In the present study, we report the presence of TaSi<sub>2</sub> peaks at 600°C. This may be due to various factors such as the integrity of the TaSi<sub>2</sub> sputter target, film thickness, and the characteristics of Si substrates.

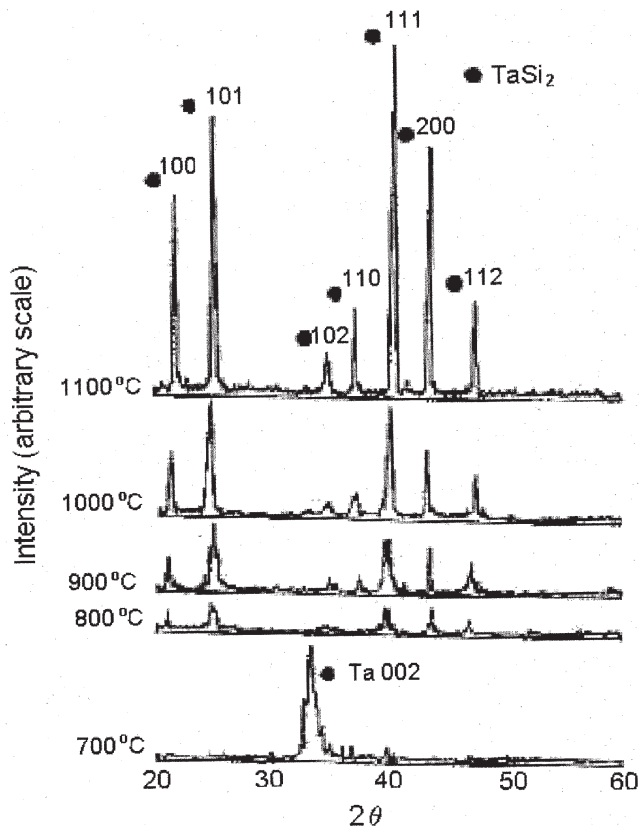
#### XRD Measurements on TaSi<sub>2</sub> Films after Wet Oxidation

According to the ternary Ta-Si-O phase diagram shown in Fig. 7, it can be predicted that the phase connected by a solid tie-line is stable. Thus, Ta<sub>2</sub>O<sub>5</sub> can be produced by the oxidation of TaSi<sub>2</sub> at temperatures in the range of 600–1100°C. A comparison of our XRD data with Jiang's<sup>8</sup> of Ta<sub>2</sub>O<sub>5</sub> thin films in Fig. 8 shows that there is no Ta<sub>2</sub>O<sub>5</sub> phase formation after wet oxidation of TaSi<sub>2</sub> thin films. From our XRD data shown in Fig. 9, no visible tantalum oxide peaks are found. All the peaks obtained in this spectrum are peaks that are comparable to the TaSi<sub>2</sub> data in Fig. 4. Based on these experimental observations, it can be concluded that there is no oxidation occurring in TaSi<sub>2</sub> thin films that are prepared using high-purity TaSi<sub>2</sub> sputter targets.

Murarka et al.<sup>9</sup> have reported their studies on TaSi<sub>2</sub> films formed on polysilicon by sintering tantalum thin films on doped polysilicon. They find that these TaSi<sub>2</sub> films do not oxidize in dry O<sub>2</sub> and oxidize in steam, giving SiO<sub>2</sub> and a surface layer of mixed Ta-Si oxides. During wet oxidation of TaSi<sub>2</sub>, the top oxide layers do not act as a diffusion barrier for water molecules. On continued oxidation, SiO<sub>2</sub> is formed, and silicon in the silicon-rich layer is replenished continuously by steady diffusion of silicon from polycrystalline silicon to the silicide-SiO<sub>2</sub> in-

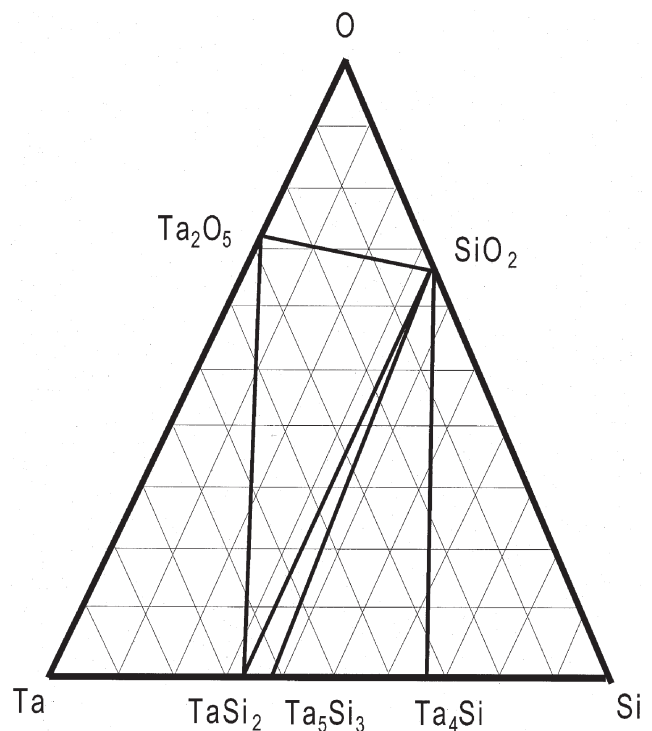
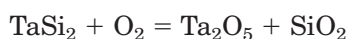
Table II. XRD Data of TaSi<sub>2</sub> (101) Peaks for Different Annealing Temperatures

Annealing Temperature (°C)	2-Theta (°)	D-Spacing (Å)	Intensity	FWHM
600	24.60	3.6259	124	2.594
700	24.680	3.6044	135	1.187
800	24.760	3.5929	187	0.836
900	24.640	3.6101	887	0.312

Fig. 6. XRD spectra of TaSi<sub>2</sub>/Si (Work of Cho<sup>3</sup>).

interface. This steady flow is established during the early stages of oxidation. No further oxidation of tantalum takes place because of the shielding effect of silicon-rich layer. This is schematically shown in Fig. 10. We have also performed XRD measurements on TaSi<sub>2</sub> (100 Å and 900 Å) from 2θ = 30° to 60°. From our XRD data analyses for the 100 Å and 900 Å films, we were not able to observe any phases of Ta<sub>x</sub>O<sub>y</sub> or SiO<sub>2</sub>.

Saraswat et al.<sup>10</sup> have reported that when TaSi<sub>2</sub> is deposited by simultaneous sputtering of tantalum and silicon on SiO<sub>2</sub>/Si, a mixture of SiO<sub>2</sub> and Ta<sub>2</sub>O<sub>5</sub> is formed. However, in the case of TaSi<sub>2</sub> films deposited on silicon, only TaSi<sub>2</sub> diffraction peaks are observed. The results of their XRD spectra are shown in Fig. 11. The schematic summarizing their results is shown in Fig. 12. In this case, the reaction of TaSi<sub>2</sub> with O<sub>2</sub> can be written as

Fig. 7. Ternary Ta-Si-O phase diagram (600–1100°C).<sup>3</sup>

As has been observed in our detailed x-ray analyses, the sputter-deposited TaSi<sub>2</sub> films are stable when heated in wet oxygen. This is summarized schematically in Fig. 13.

## CONCLUSIONS

TaSi<sub>2</sub> thin films were sputter deposited on p-Si and n-Si wafers by using a high-purity TaSi<sub>2</sub> target. The thicknesses of TaSi<sub>2</sub> thin films were in the range 100 Å to 1000 Å. The wafers were then annealed at temperatures in the range of 400°C to 900°C in nitrogen, forming gas, or oxygen containing steam. Four-point probe results show that the sheet resistance of the TaSi<sub>2</sub> thin film decreases with an increase in film thickness and annealing temperature. X-ray diffraction results show that the microstructure of TaSi<sub>2</sub> thin film changes from amorphous to crystalline as annealing temperature increases. In the annealing temperature range of 800°C to 900°C, the characteristics of diffraction peaks change considerably. Wet oxidation characteristics of TaSi<sub>2</sub> thin films have been investigated in the temperature range of 400–900°C. According

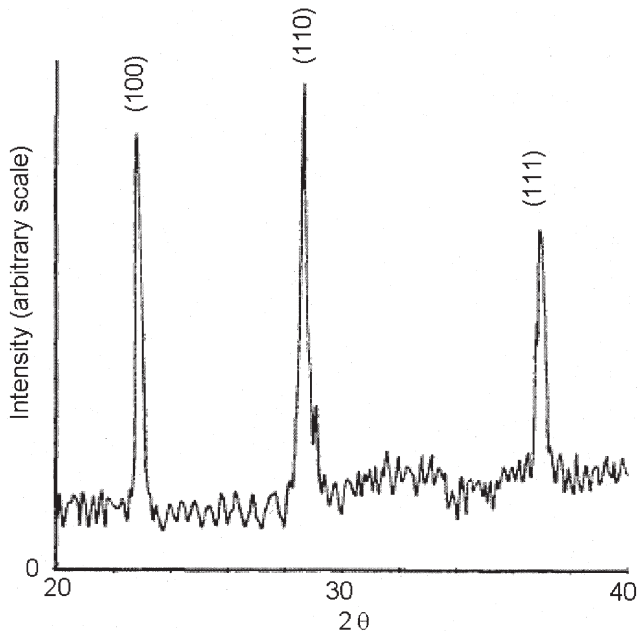


Fig. 8. XRD spectra of Ta<sub>2</sub>O<sub>5</sub> thin films.

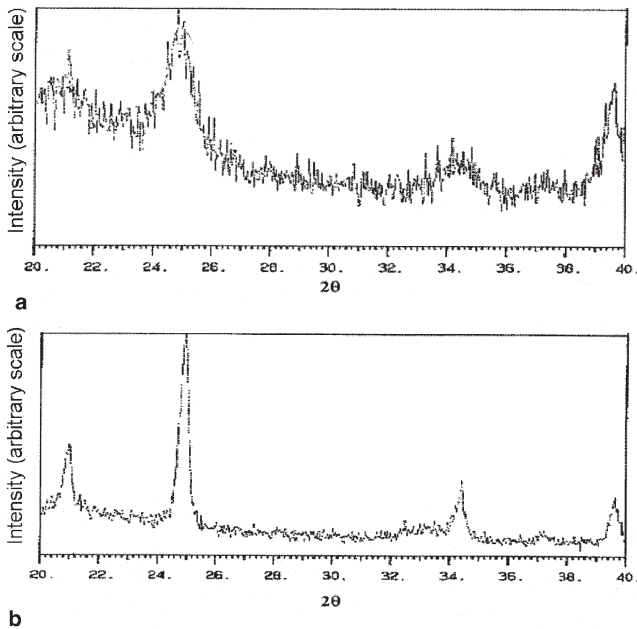


Fig. 9. XRD spectra after wet oxidation of 1000 Å TaSi<sub>2</sub> thin film on n-Si: (a) 700°C for 1 h and (b) 900°C for 1.5 h.

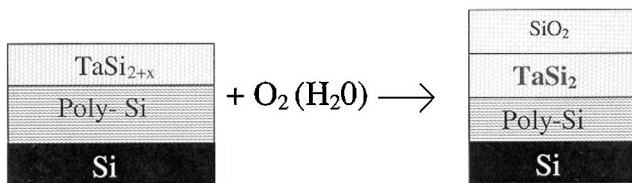


Fig. 10. Schematic of wet oxidation of TaSi<sub>2</sub> films formed on polysilicon by sintering tantalum film on doped poly.<sup>9</sup>

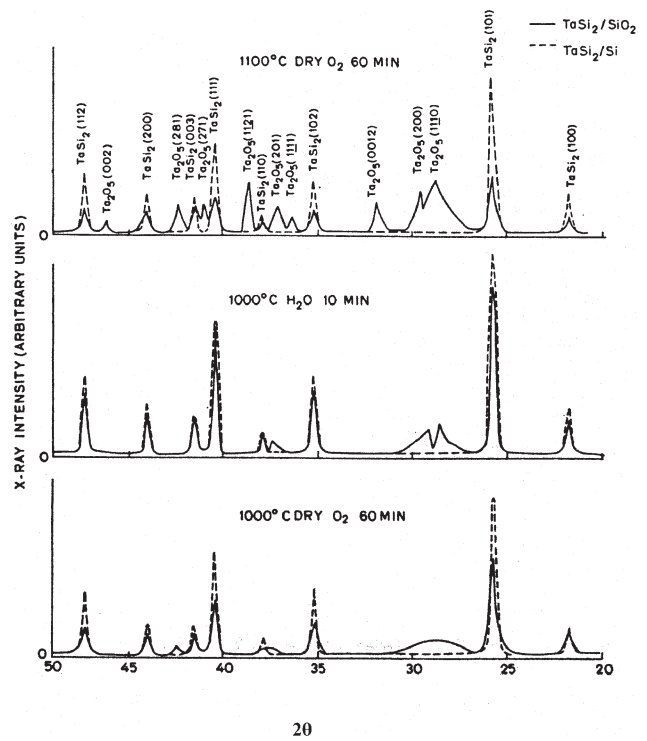


Fig. 11. XRD of oxidation of tantalum silicide films.<sup>10</sup>

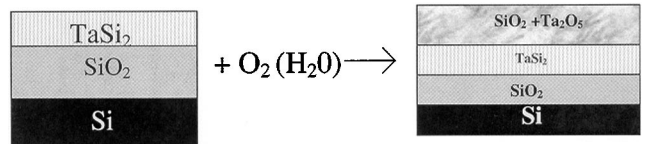


Fig. 12. Schematic showing oxidation of TaSi<sub>2</sub> films deposited by simultaneous sputtering of tantalum and silicon. In the case of TaSi<sub>2</sub>/Si samples, only TaSi<sub>2</sub> diffraction peaks are observed, whereas for TaSi<sub>2</sub>/SiO<sub>2</sub> samples, additional diffraction peaks of Ta<sub>2</sub>O<sub>5</sub> are also observed.<sup>10</sup>

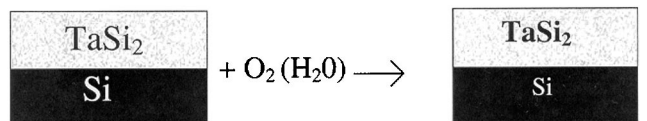


Fig. 13. Schematic showing wet oxidation of sputter-deposited TaSi<sub>2</sub> films using ultrapure TaSi<sub>2</sub> target.

to our XRD results, there are no detectable Ta<sub>2</sub>O<sub>5</sub> peaks in the films.

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